

# Carbon and lithium sputtering in the NSTX divertor

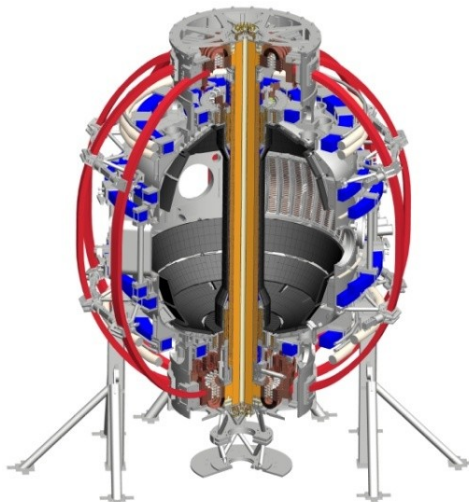
Filippo Scotti

Acknowledgements:

V.A. Soukhanovskii, J-W. Ahn, M.Jaworski, T.Gray

**NSTX Monday Physics Meeting  
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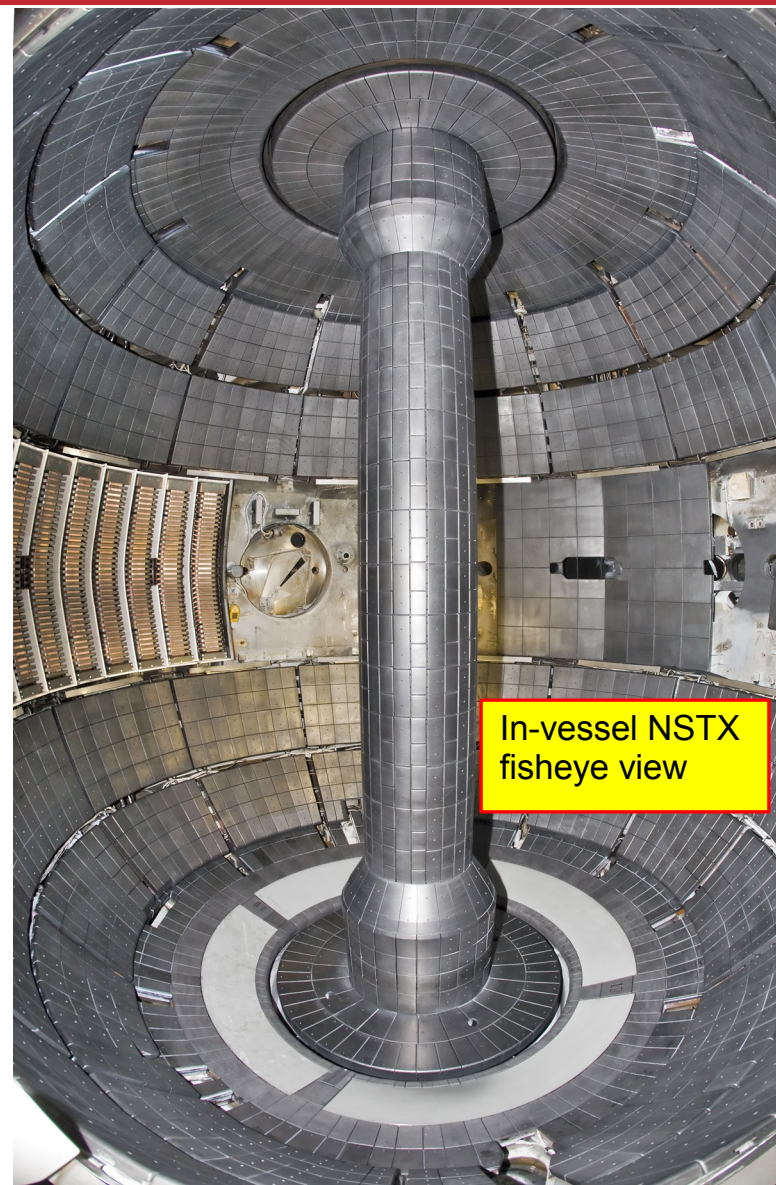
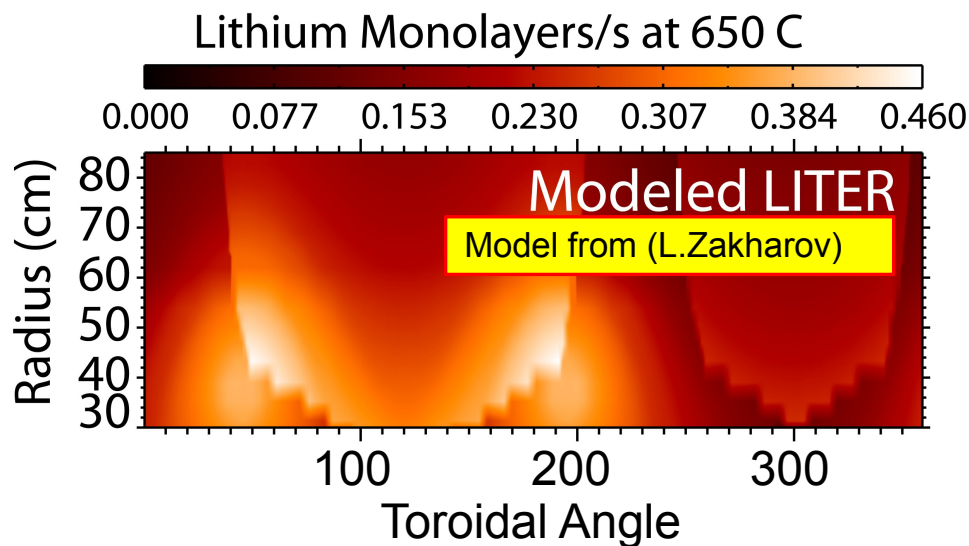
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# Outline

- NSTX plasma facing components (PFCs)
- Divertor diagnostics and analysis method
- Expected behavior of lithium and carbon sputtering
- Lithium sputtering in NSTX divertor
- Carbon sputtering in NSTX divertor
- Conclusions and future work

# Evaporative lithium coatings are routinely applied on NSTX graphite and moly PFCs as wall conditioning technique

- Graphite is the main PFC material in NSTX:
  - ATJ graphite tiles on divertor and main wall
  - ATJ and CFC tiles on center stack
  - Porous moly segments in outer divertor (LLD)
- Lithium coatings evaporated on PFCs:
  - 100 – 300 mg of lithium are typically applied between discharges using LITERs
  - “coating” thickness ~ few 10s of nm
- Highly toroidally asymmetric deposition

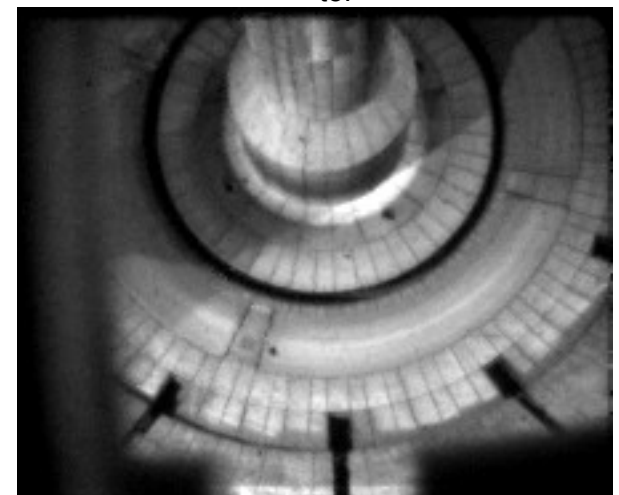
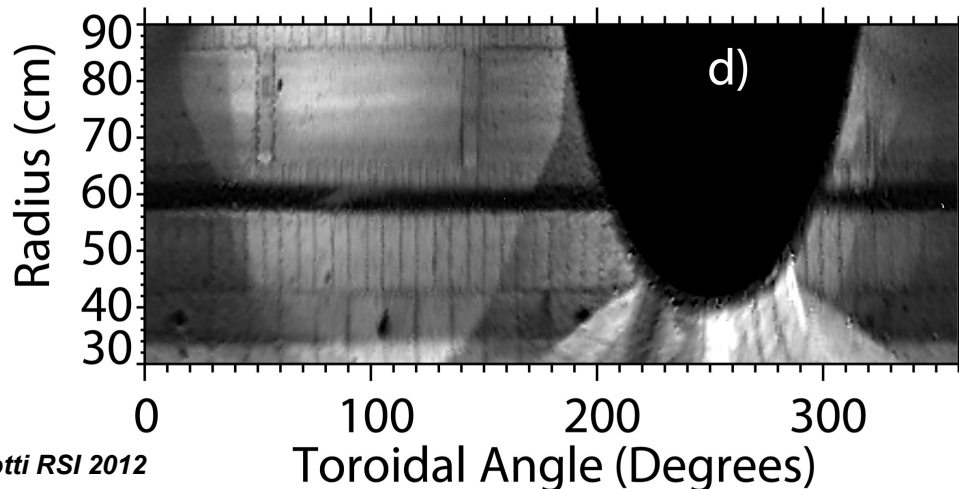


# Divertor cameras, high resolution spectrometer, Langmuir probes used for analysis of impurity influxes

- Spectroscopic diagnostics for divertor impurity influxes:
  - VIPS2, 1D/2D cameras (Li I 670nm, Li II 548nm, C II 658nm, 392nm, C III 465nm)
- Langmuir probes for  $T_e$ ,  $n_e$ ,  $J_{sat}$ :
  - inboard (LP2913, R=49.5 cm), outboard HDLP array (4 single, 2 triple)
- Two color IR thermography for divertor  $T_{surf}$
- Inverse photon efficiencies (S/XB) from ADAS
- Sputtering yield derived from impurity influxes:

$$\Gamma_i = \frac{S}{XB} \Gamma_{ph} \quad Y_I = \frac{\Gamma_I}{\Gamma_{D+}} = \frac{B_I \cdot S / XB}{J_{SAT}}$$

- 2D camera data remapped in  $(r, \Phi)$  for easier analysis at  $\neq \Phi_{tor}$





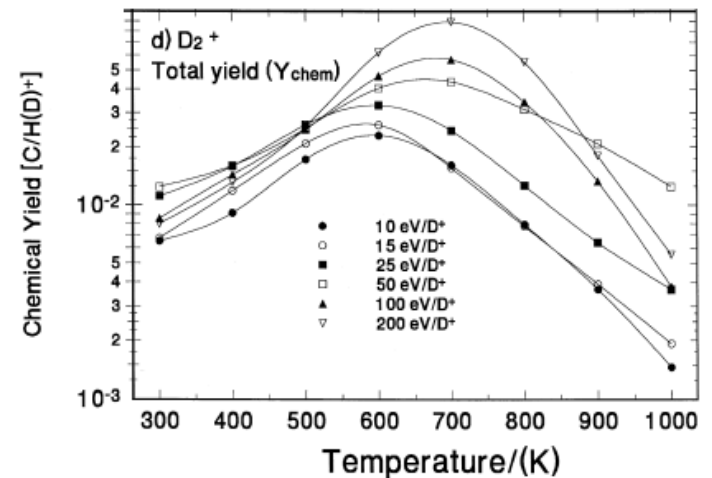
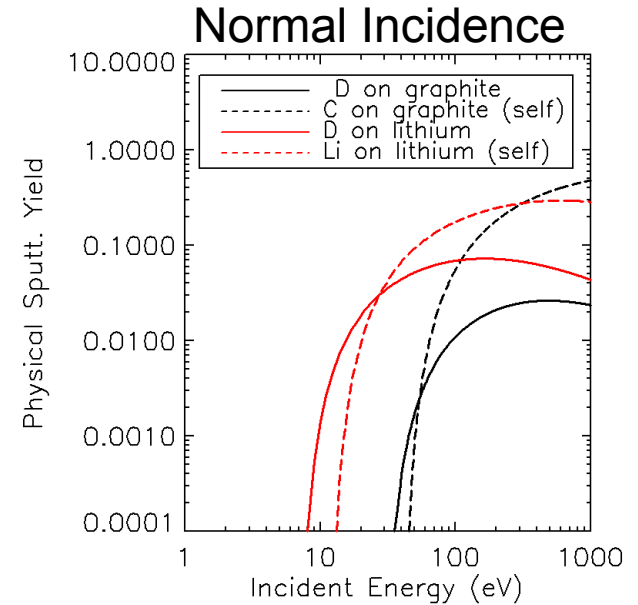
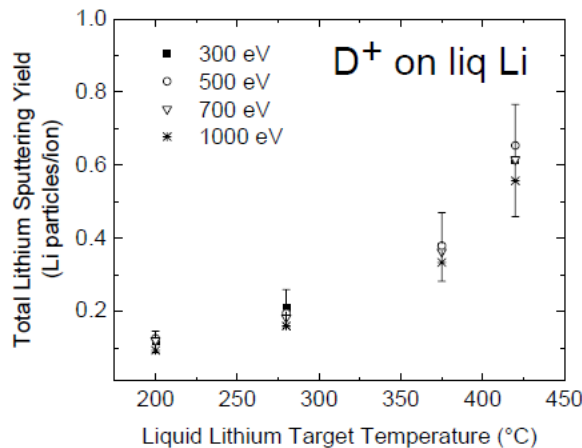
# Carbon and lithium sputtering yields show incident energy thresholds and $T_{surf}$ , $T_e$ dependence

## Carbon:

- Physical Sputtering ( $E_{th} \sim 30\text{eV}$ )
- Chemical Sputtering (no  $E_{th}$  but  $T_{surf}$  dep.)
- Sources expectations: ion impact on divertor and main wall, neutrals on first wall and divertor

## Lithium:

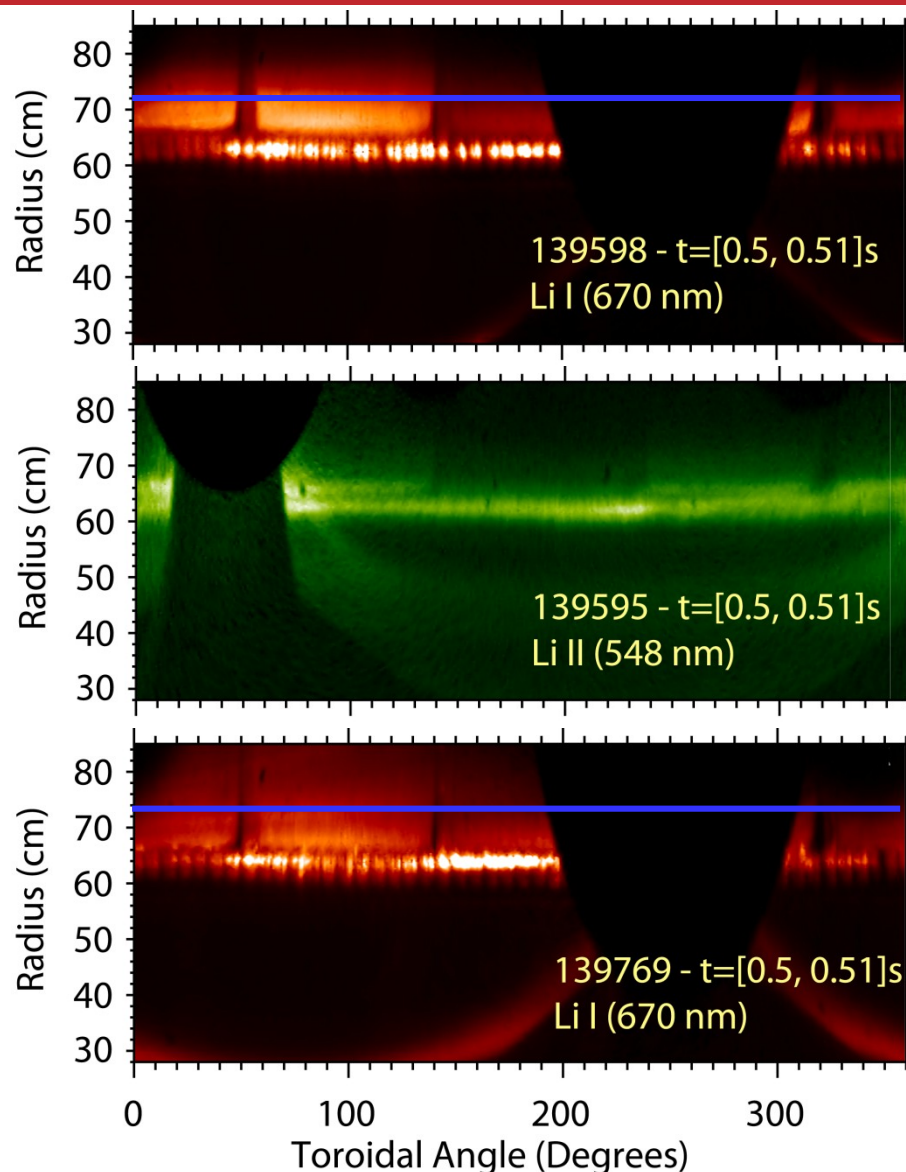
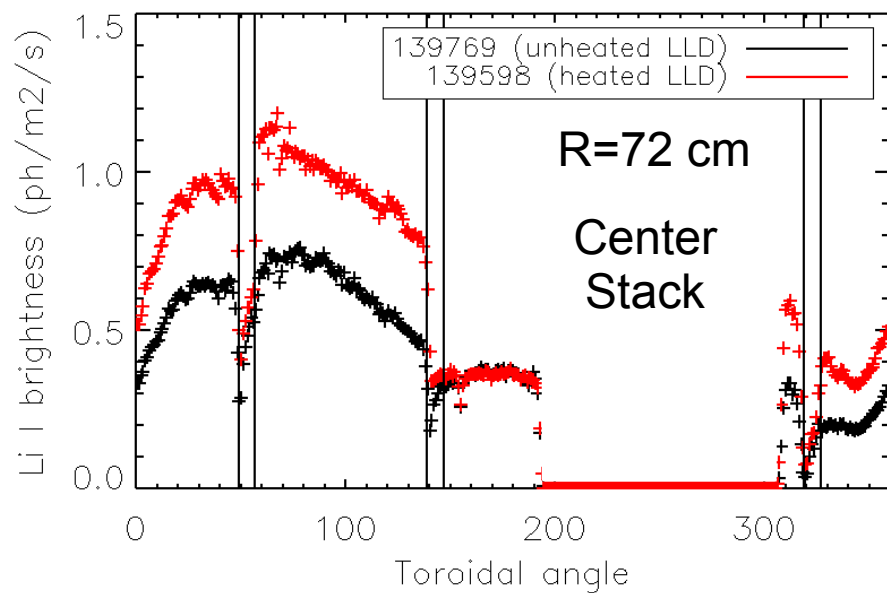
- Physical Sputtering only ( $E_{th} \sim 7\text{eV}$ )
- Strong  $T_{surf}$  dependence
- 2/3 of the particles are sputtered as ions
- Sources expectations: ion impact on divertor



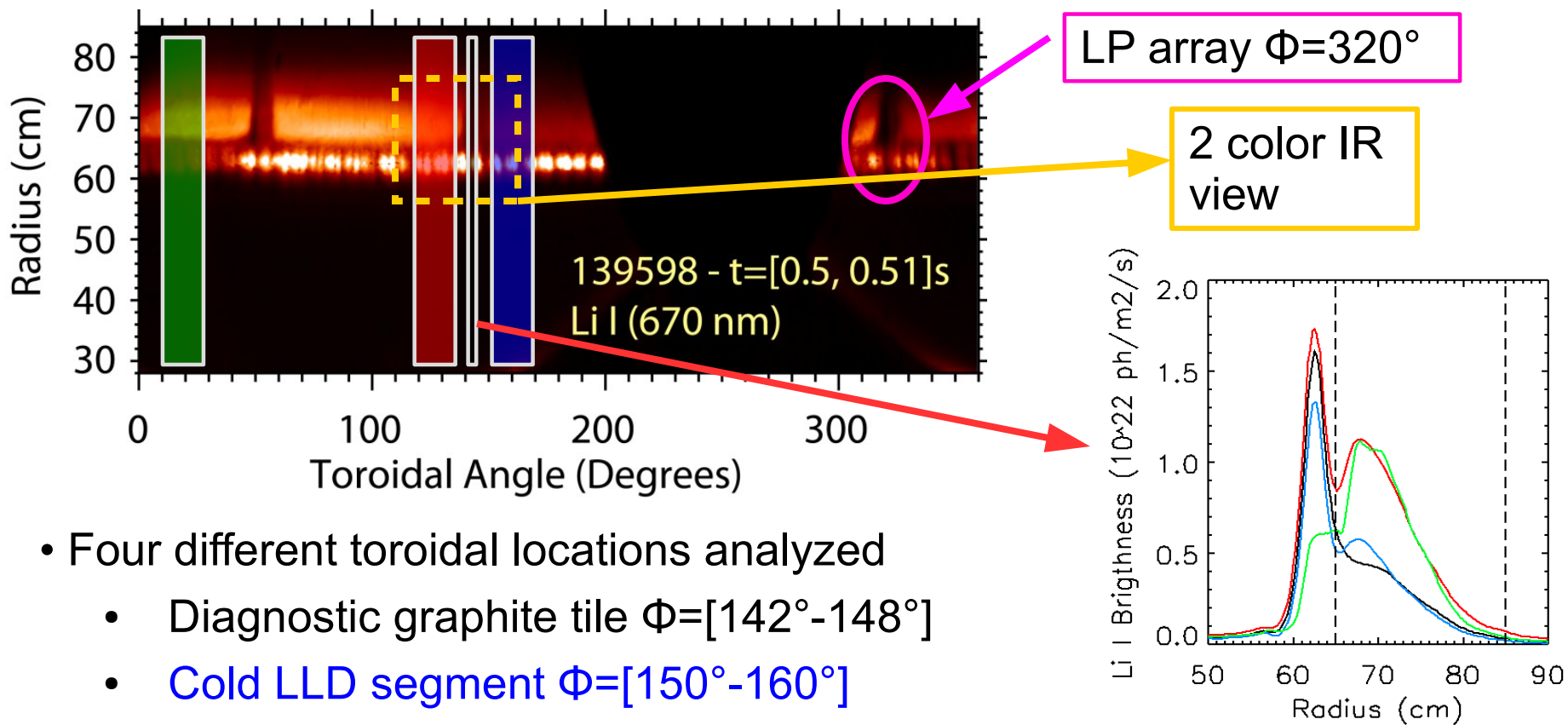
B.V. Mech et al. J. Nucl. Mater. 255, 153–164 (1998).

# Differential heating of plasma facing components leads to toroidal asymmetries in lithium influxes

- Low  $\delta$  discharges post mega evaporation (>200g): one with heated and one with unheated LLD (139598, 139769)
- At R= 63cm ~ coating thickness > 20  $\mu\text{m}$
- Differential heating of LLD segments results in non-axisymmetric lithium influxes
- 139598 representative of many of the Li sputtering behaviors



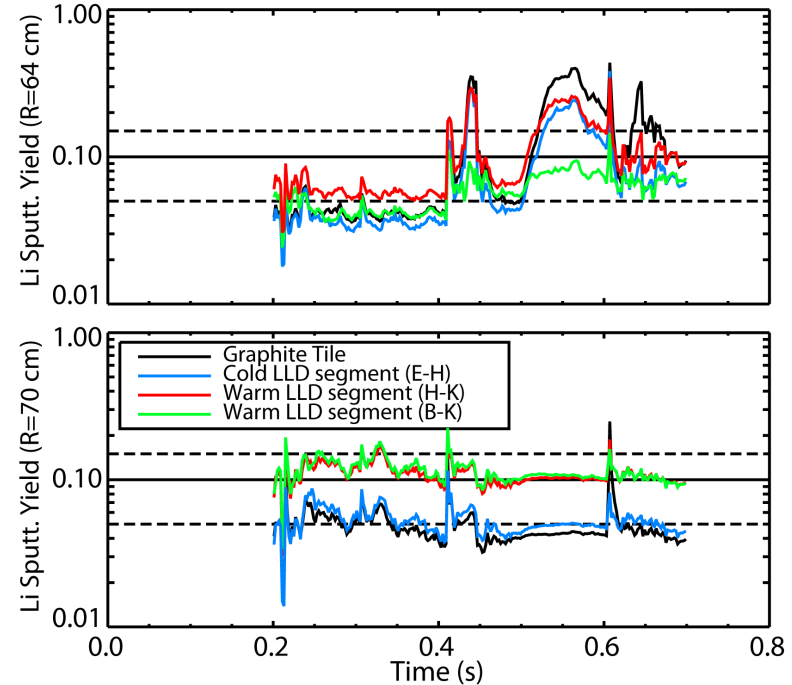
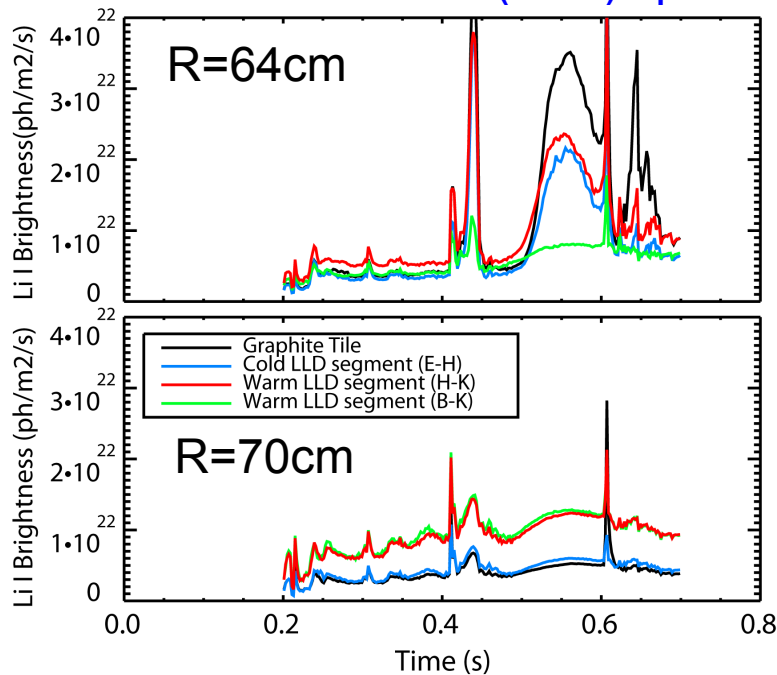
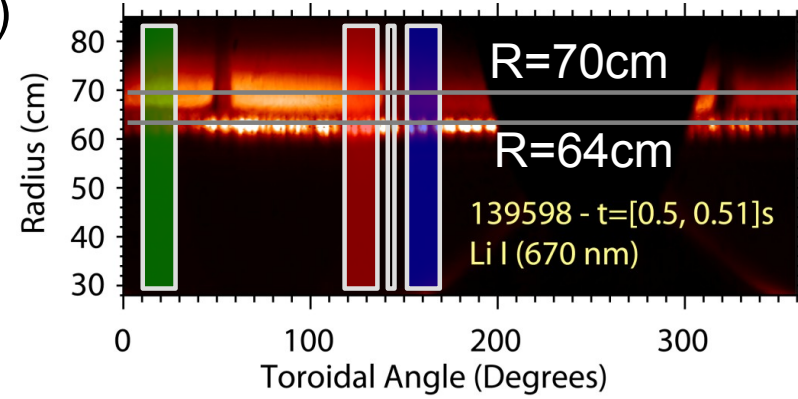
# Analysis of lithium sputtering is done using 2D cameras (at different $\Phi_{\text{tor}}$ ), HDLP array data and dual band IR



- Four different toroidal locations analyzed
  - Diagnostic graphite tile  $\Phi=[142^\circ-148^\circ]$
  - Cold LLD segment  $\Phi=[150^\circ-160^\circ]$
  - Hot LLD segment  $\Phi=[120^\circ-135^\circ]$  + thick coating location
  - Hot LLD segment  $\Phi=[10^\circ-30^\circ]$  + thin coating location
- Analysis at two radial locations will be presented (R=64, 70 cm) @ single probes

# Analysis of different toroidal locations shows evidence of Li temperature enhanced sputtering

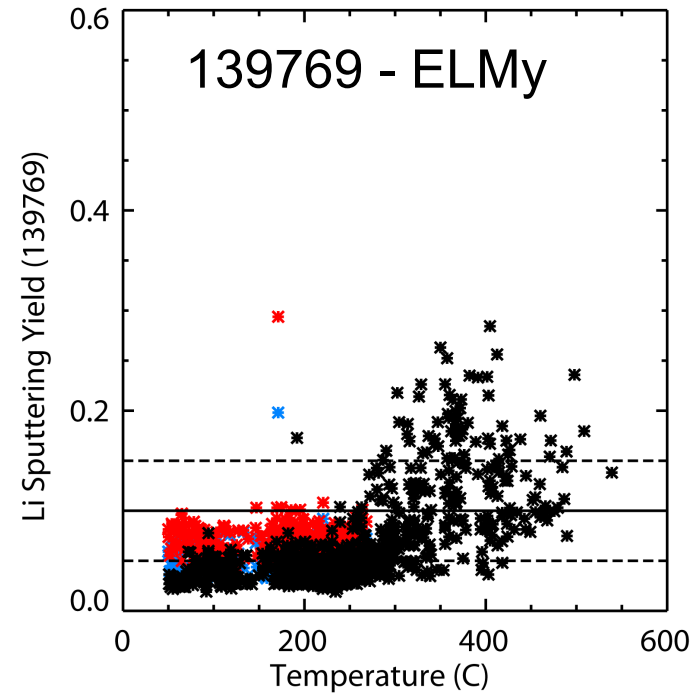
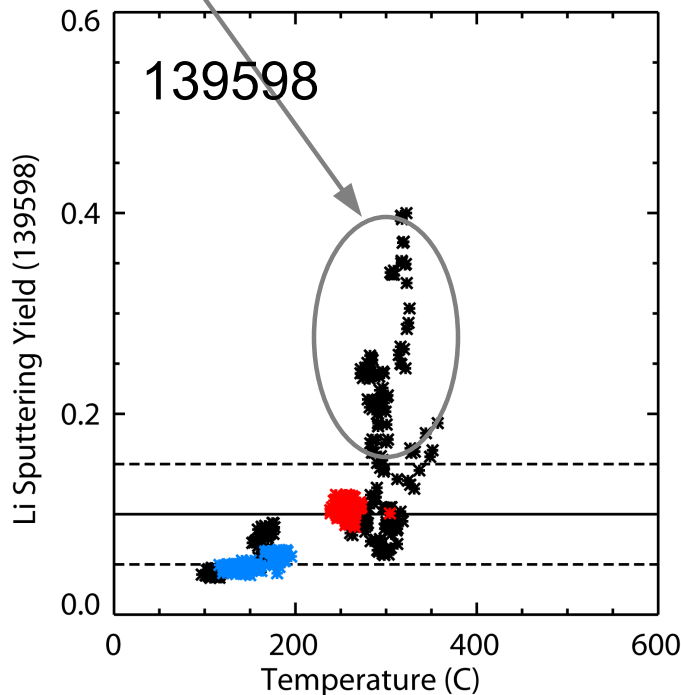
- $T_{\text{surf}}$  dependence of sputtering yield (up to 2X)
- Thick coatings  $\rightarrow$  same response from moly and graphite
- Evidence of Li further enhanced sputtering
  - Non linearity in yield at strike point, observed only in thicker deposition areas
  - Related to self (or C) sputtering runaway?





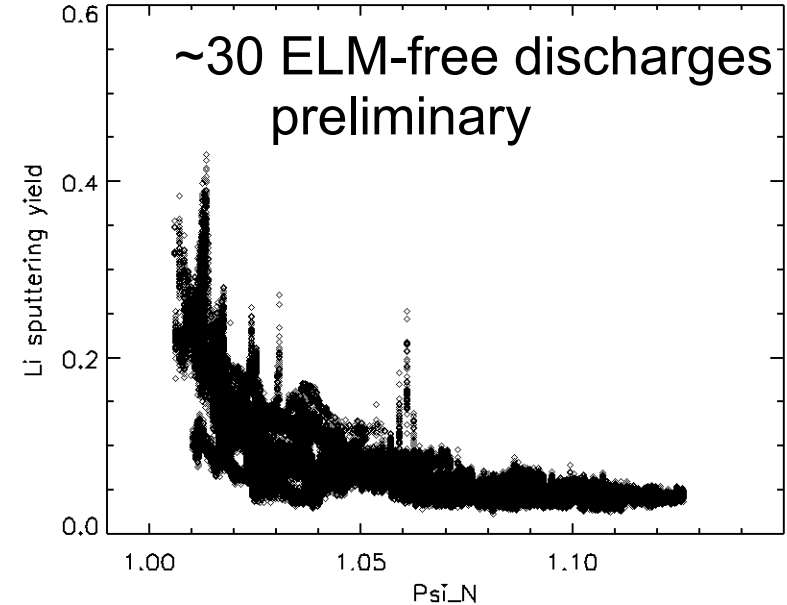
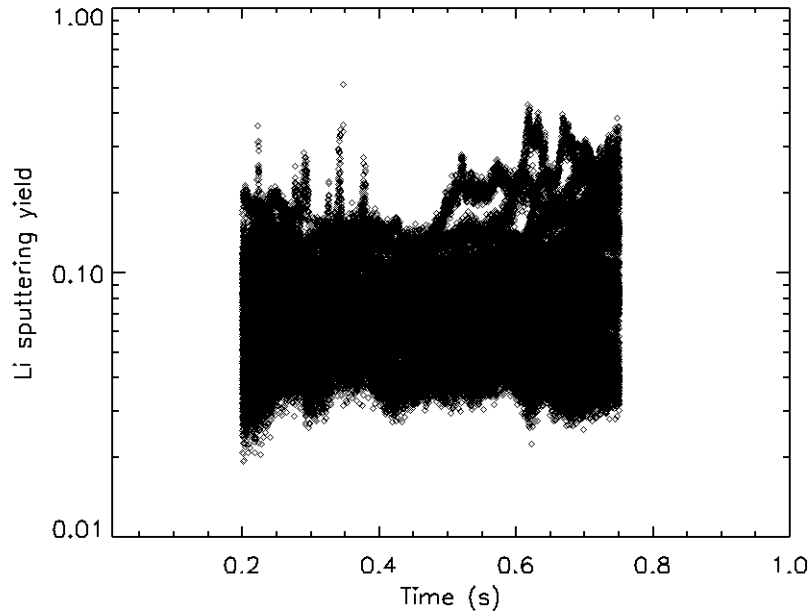
# Measured lithium sputtering yield consistent with literature with signature of temperature dependent sputtering yield

- Using local  $T_{\text{surf}}$  from 2-color IR, dependence of sputtering yield on  $T_{\text{surf}}$
- Assuming  $T_{\text{surf}}$  is accurate, evaporation flux is negligible
- For Li bloom,  $T_{\text{surf}}$  seems to be clamped at OSP at  $\sim 300\text{C}$ 
  - Can't be explained by evaporation or  $T_{\text{surf}}$  dependence only (at  $350\text{C}$  evaporation rate  $\sim 5e19$  atoms/m<sup>2</sup>/s)



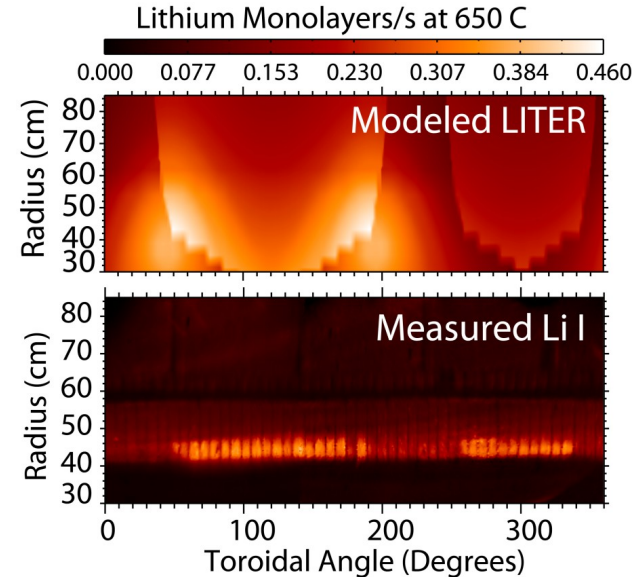
# Preliminary statistical analysis carried out on graphite (high $\delta$ discharges) also suggests $T_{\text{surf}}$ dependent lithium sputtering

- Large variability in Li sputtering observed on LP2913 probe (R=49cm)
- No apparent decrease over time
- Plotting Yield vs  $\Psi_N$  (natural strike point sweep) reduces variability and suggest  $T_{\text{surf}}$  dependence
- Some impact of pre-applied Li amount, PFC history likely to play a role

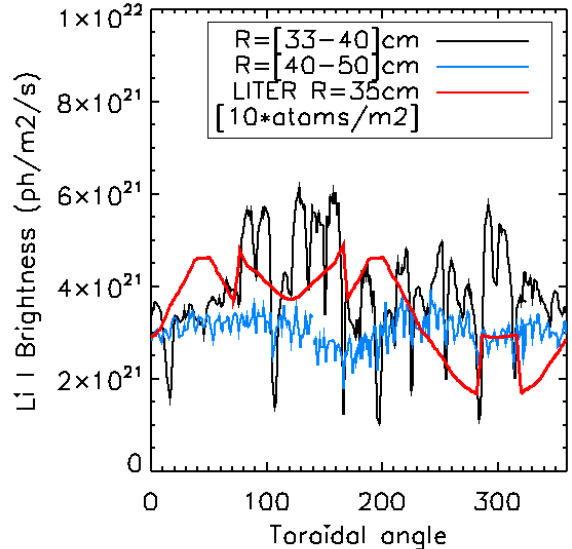


# Effect of different amount of lithium in high $\delta$ discharges studied analyzing different toroidal locations in same discharge

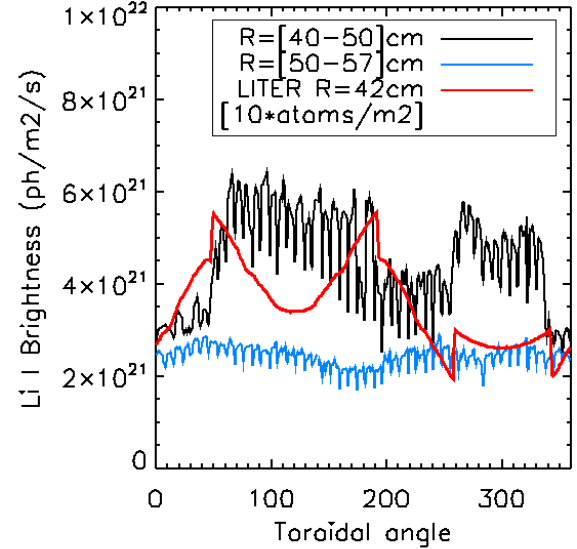
- High  $\delta$  discharges at  $\neq \Phi_{tor}$
- No  $\neq$  in Li yield in far SOL at  $\neq \Phi_{tor}$ 
  - Not expected if Li thickness  $>$  ion range
- Differences (up to 2x) observed at OSP, enhanced during transients (e.g. ELMs)
  - Qualitative agreement with LITER pattern
  - Consistent with enhanced response from thicker coatings areas (similar to Li “bloom”)



139013  
 $t=[0.4, 0.45]$   
 ELM free

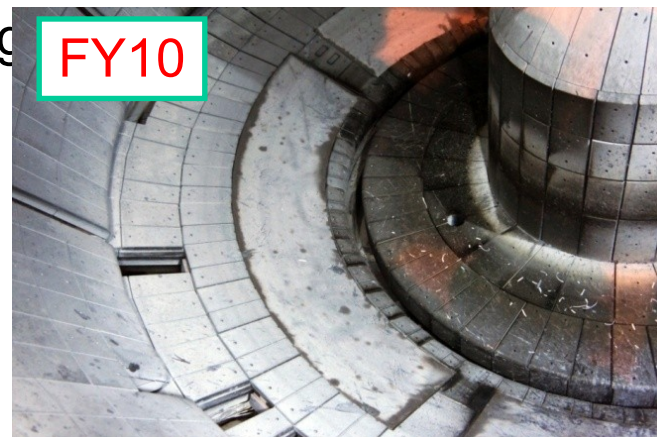


139027  
 $t=[0.4, 0.41]$   
 during ELM

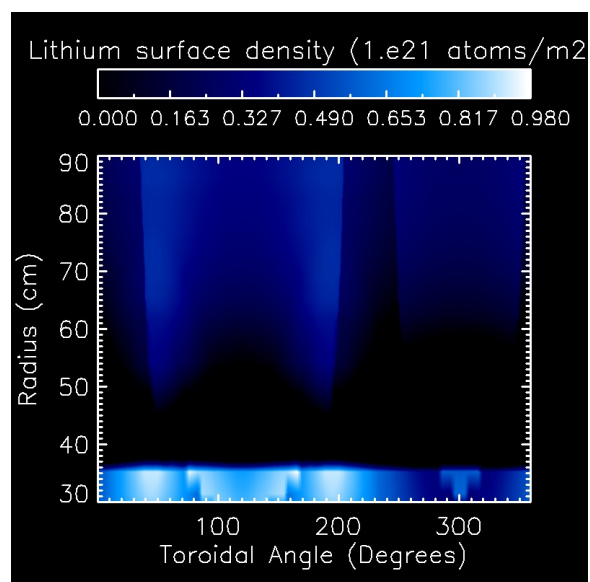
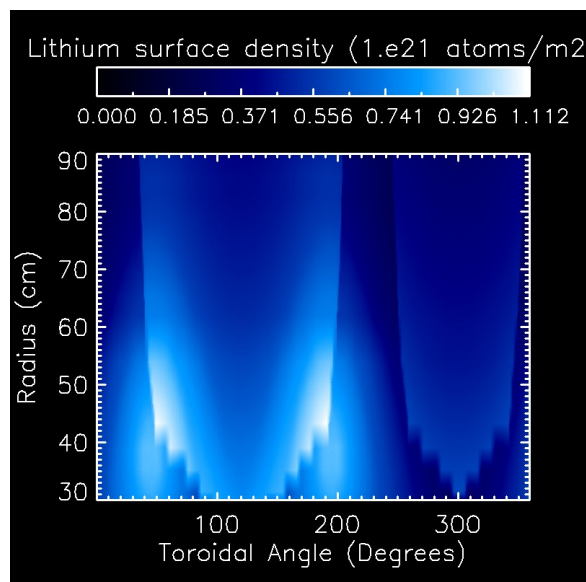


# Degradation of coatings is observed from post vent visual inspection and nuclear analysis of divertor tiles

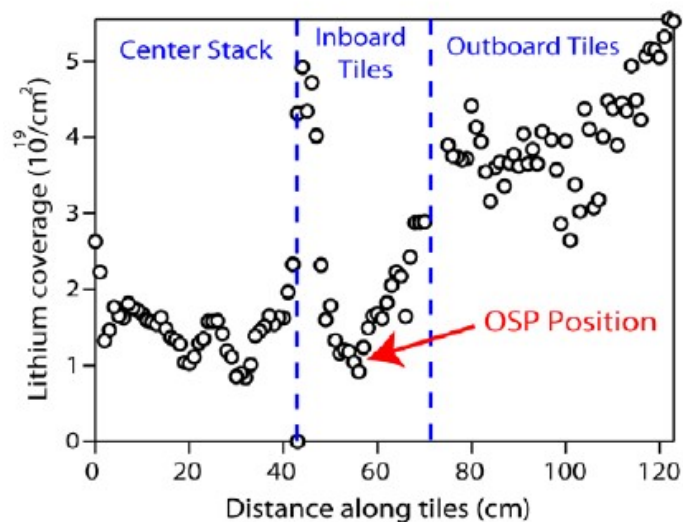
- Fresh coatings thickness ~thicker than  $D^+$  stopping range (15-45 nm for 200 mg Li)
- Coating can be eroded from incoming ion flux
- Important role of prompt re-deposition
  - No clear OSP lithium depletion within gross erosion timescales
- Chemistry: Li intercalation, Li reacting with C and vacuum gases



200mg Li,  $4e23$  ion/m<sup>2</sup> incident,  $Y_{Li} \sim 0.1$ ,  $\Delta t = 0.4s$



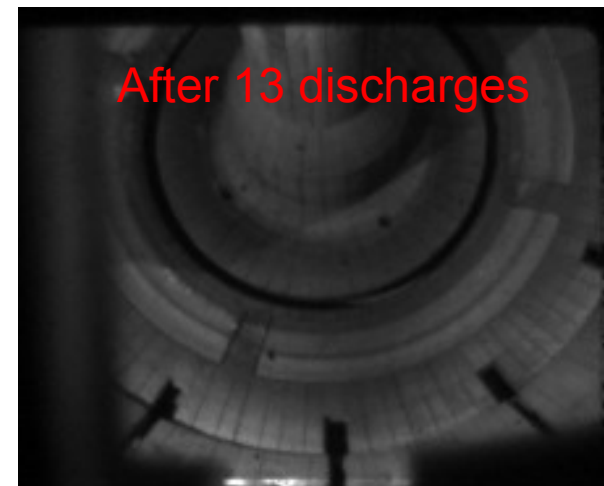
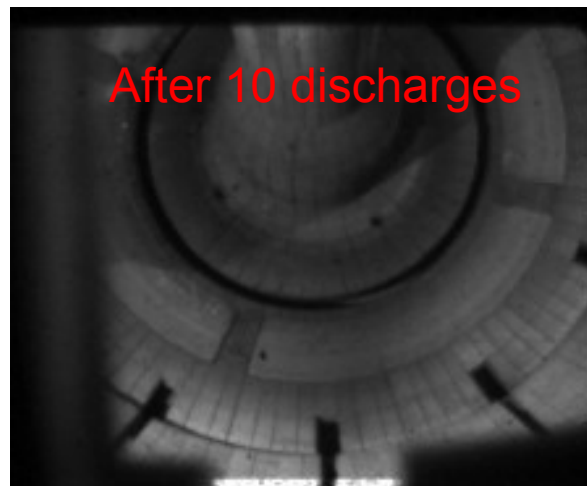
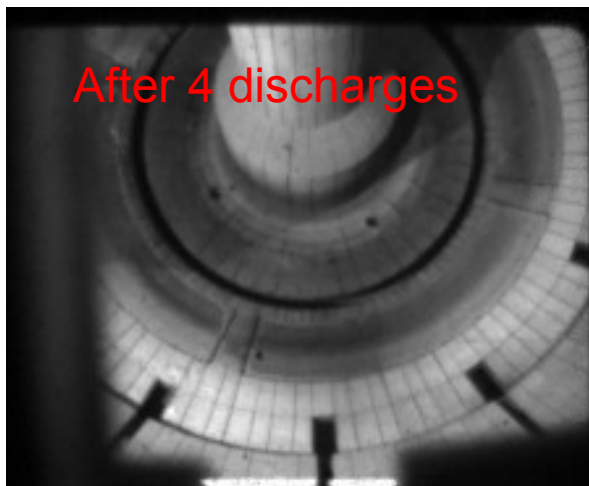
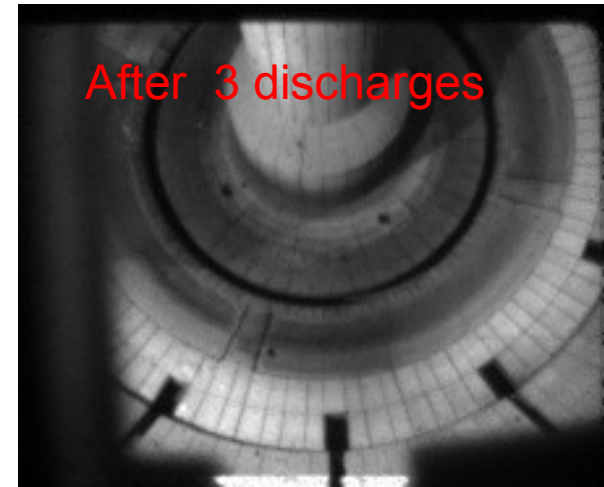
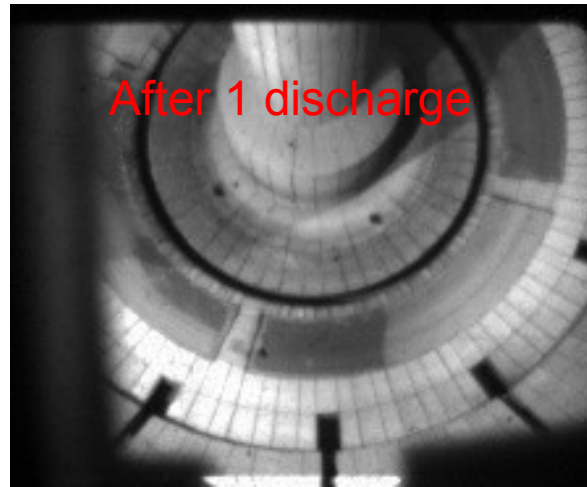
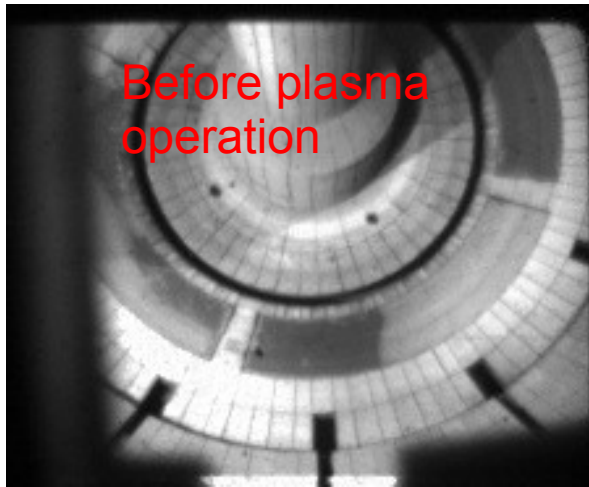
Post-Run Divertor Tiles Li coverage (Bay F, 2009)



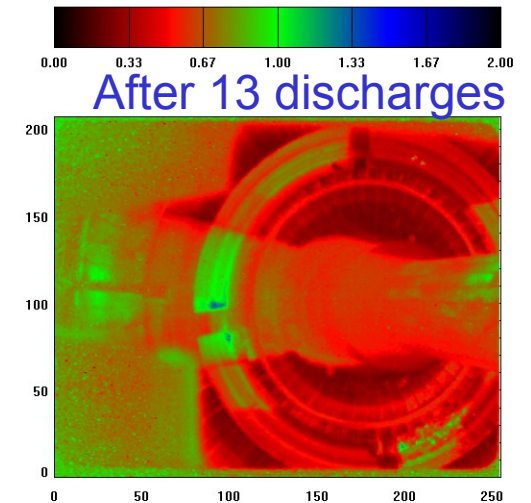
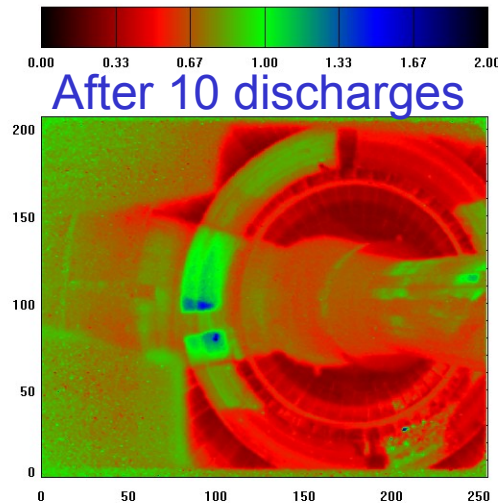
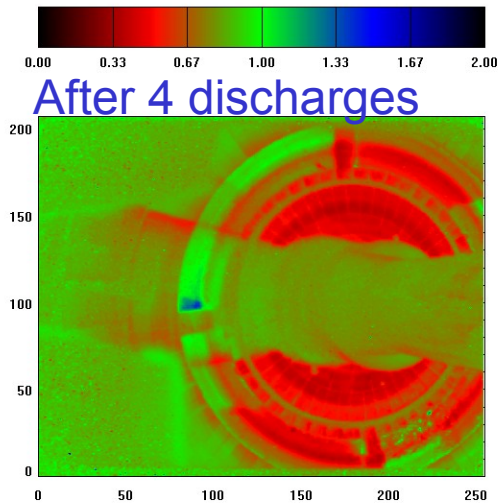
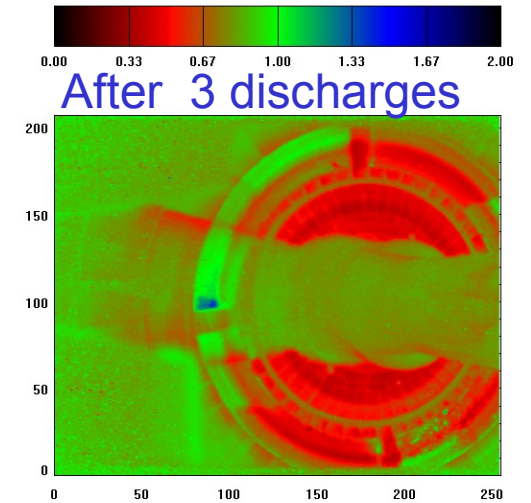
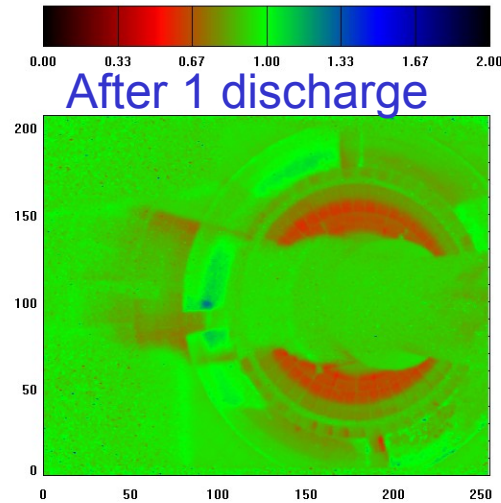
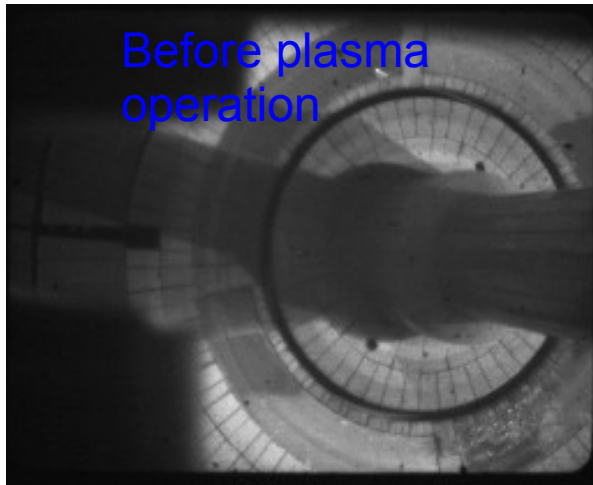
Data courtesy of W. W. Wampler (SANDIA)



# Change in surface appearance in discharges after large evaporation (6g) indicates degradation of coatings



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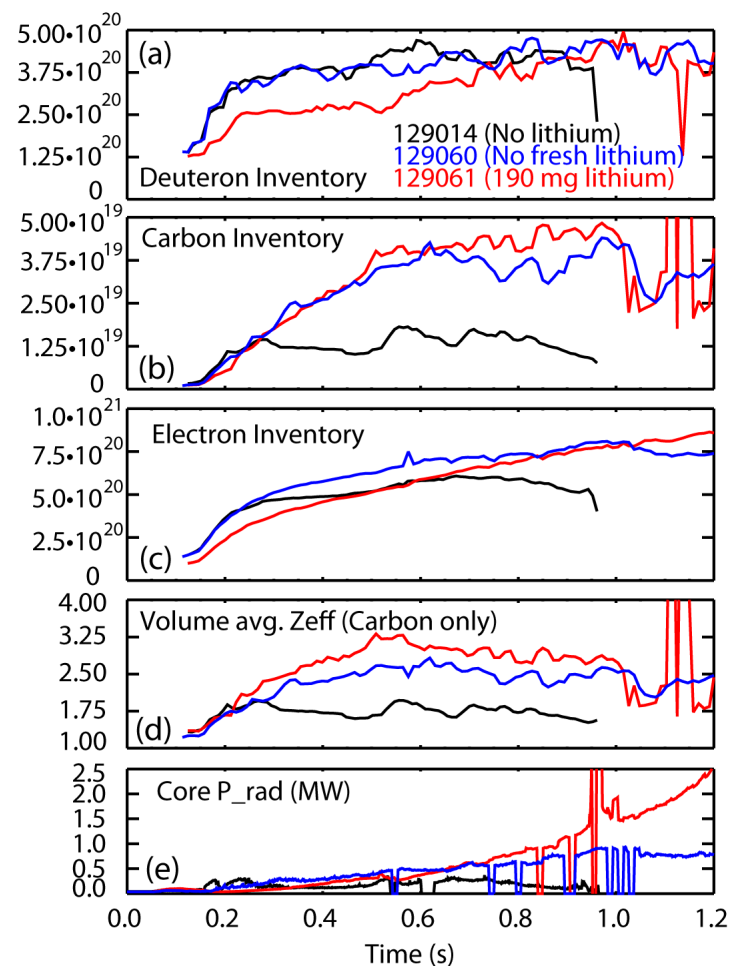


# Summary of observed lithium sputtering behavior

- Evidence for lithium temperature dependent sputtering
  - From differential heating of LLD plates
  - From radial dependence of derived sputtering yields
- Consistent with published Li sputtering (IIAX, DIII-D) but uncertainties:
  - Surface composition: Li? LiD? LiOH?,  $\text{Li}_2\text{CO}_3$ ?
  - Importance of self sputtering and impurity sputtering
  - Angle of incidence
- Evidence for toroidal asymmetry of influxes and coatings
- Evidence for deterioration of coating
  - Post vent: visible inspection and nuclear analysis (W. Wampler )
  - Visible imaging
  - Carbon influxes
  - BUT no apparent degradation of Li sputtering during discharge
- Role of prompt re-deposition under study
  - Li ionized fluxes  $\ll$  neutral influxes
  - Li coating lifetime much longer than expected from gross erosion

# Suppression of ELMs and increase in core carbon inventory with solid lithium coatings in NSTX

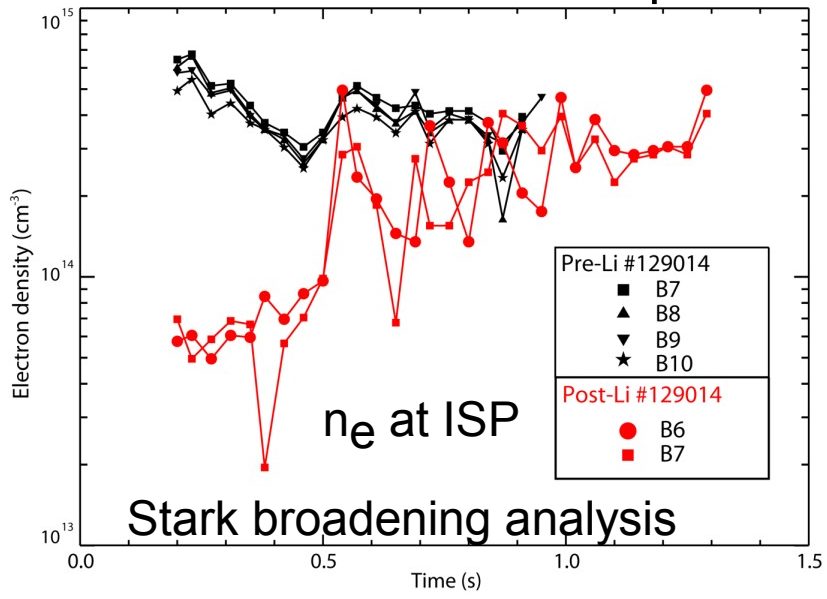
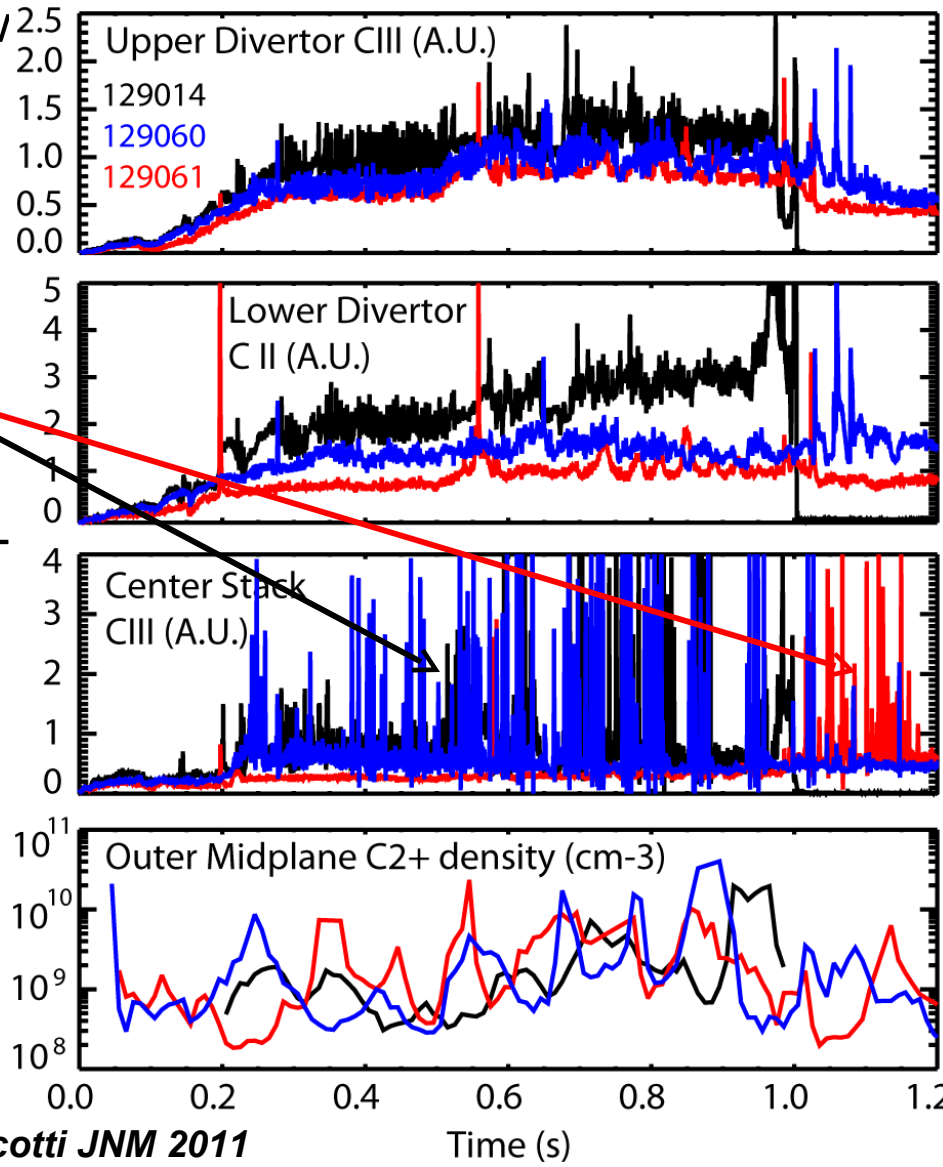
- Comparing 2008 transition from boronized graphite to lithium-coated discharges (lower single null, 4-6MW)
- Deuteron inventory reduced despite higher fueling
- Carbon inventory increased by 3-4x due to:
  - suppression of ELMs
  - changes in neoclassical inward transport
- Need to investigate role of sources and SOL transport:
  - Change in divertor/main wall carbon sources
  - Change in parallel/perpendicular transport





# Reduction of carbon emission from PFCs with application of lithium may suggest reduction of carbon influxes

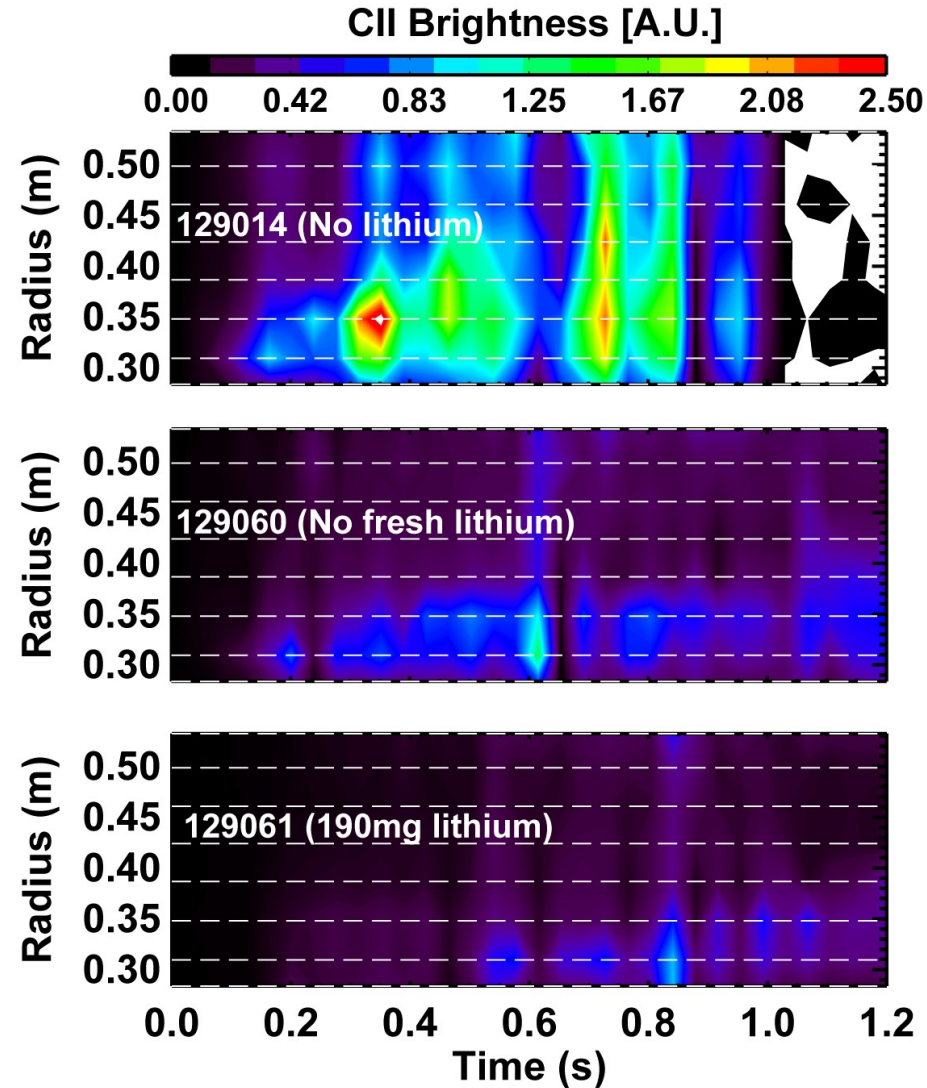
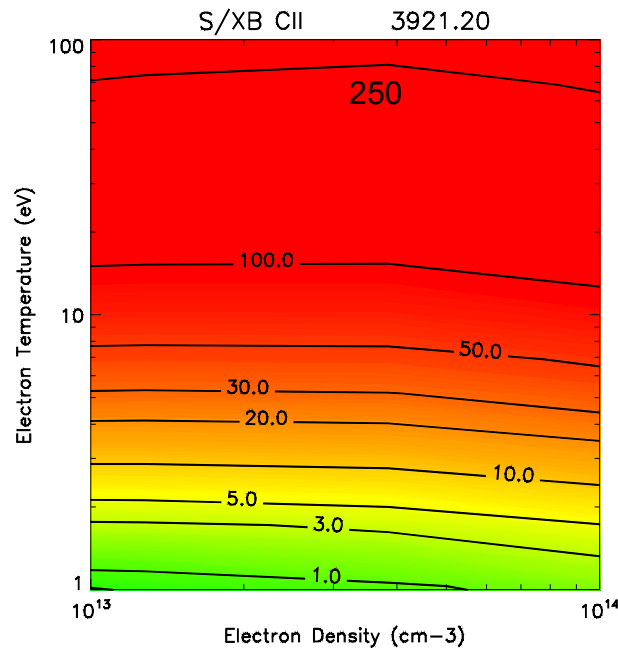
- Reduction in emission from carbon low ionization stages (C II, C III)
  - 2x in upper divertor (C III, 465 nm)
  - 3x in lower divertor (C II 658 nm)
  - 2x from center stack(C III, 465 nm)
- But  $T_e$  dependence of C II-III S/XBs !
- Disappearance of MARFE activity
  - from reduced ISP  $n_e$  and  $P_n$
- No clear trend in outer midplane  $n_{C2+}$



F. Scotti JNM 2011

# Divertor C II brightness progressively reduces with lithium application

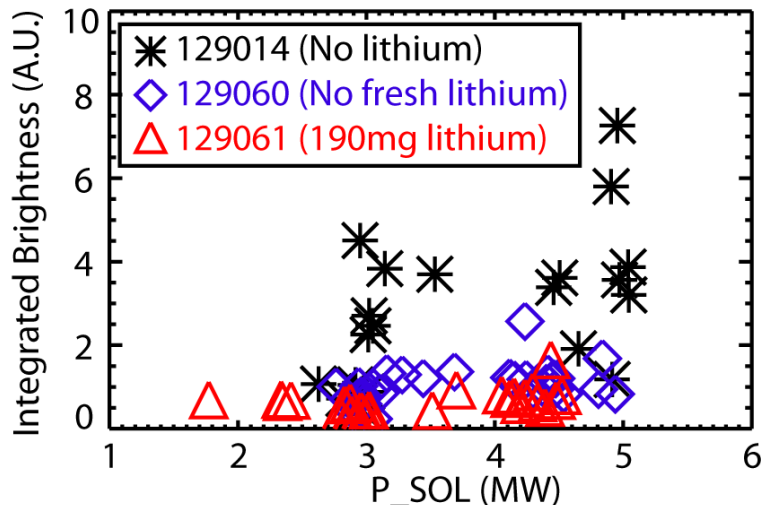
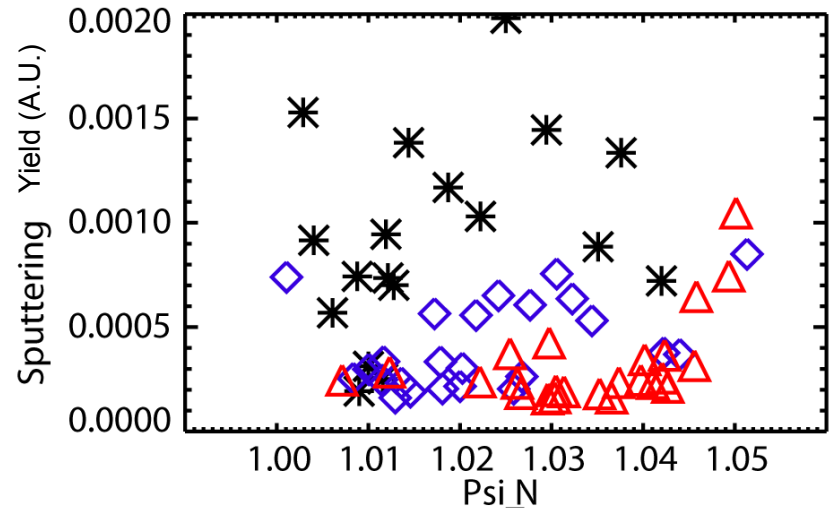
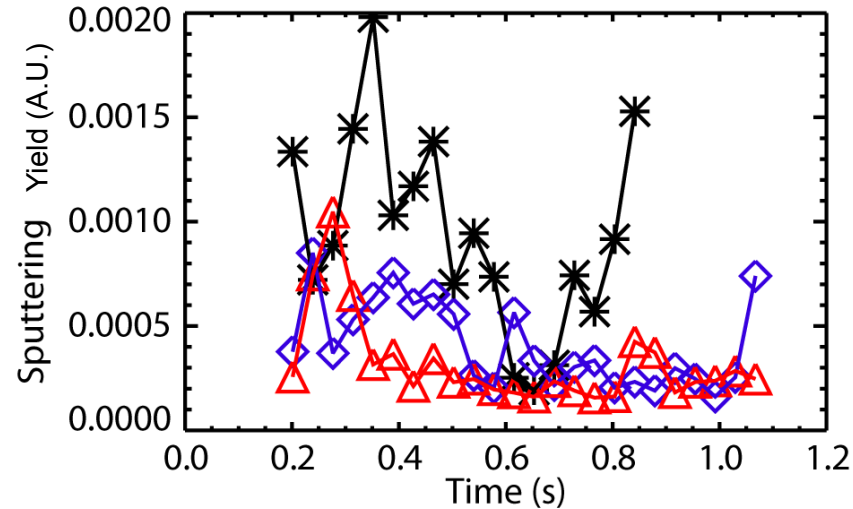
- C II brightness (392 nm) reduces in both ISP and OSP with lithium application
- Viewing chord at R=49.5 cm used to estimate changes in sputtering
- OSP integrated brightness used for change in OSP influxes
- Probe  $T_e$  and  $n_e$  used to infer S/XB factors



# Moderate reduction in sputtering rate in near/far SOL observed with application of lithium coatings

- Sputtering rate reduced with lithium
- Intermediate conditions (e.g. 129060) hard to distinguish
- Conclusions on total OSP influx limited by lack of divertor  $T_e$ ,  $n_e$  coverage
- Using the largest  $T_e$  difference measured by the probes, OSP influx is still reduced

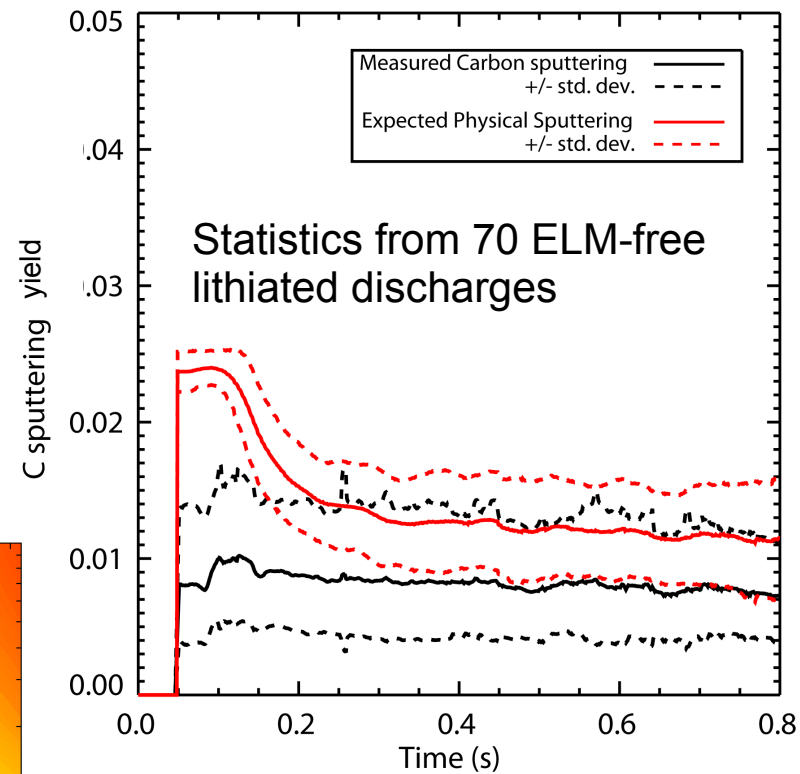
$$Y_C = \frac{J_{CII}}{J_{D^+}} = \frac{B^{@probe}_{CII} \cdot S}{XB J_{SAT}}$$



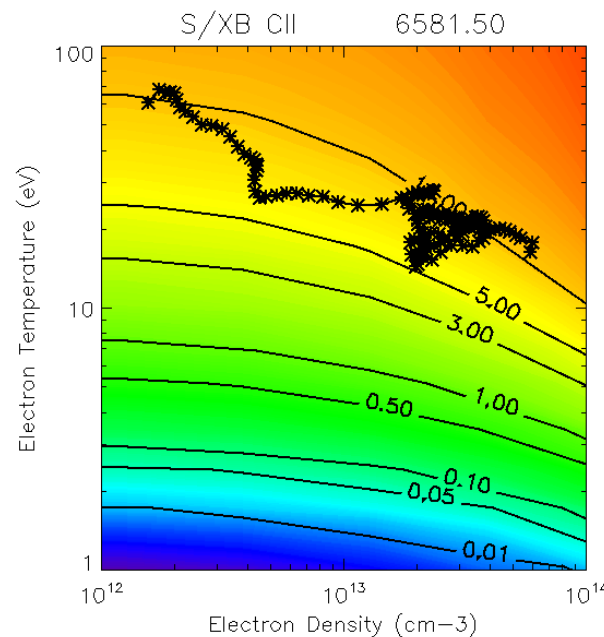
F. Scotti submitted to JNM 2012

# Results consistent with more recent data showing only a moderately reduced carbon sputtering rate on lithiated PFCs

- Data from absolutely calibrated high resolution filtered cameras
- Show moderate reduction in sputtering yield
- Measured rate is comparable to physical contribution (at normal incidence)
- Only lithiated conditions available
- Chemical sputtering behavior on Li coated PFCs unknown: to be investigated



S/XB factor for 658nm line  
( $2s^2 3p^1 \rightarrow 2s^2 3s^1$ )  
not too problematic



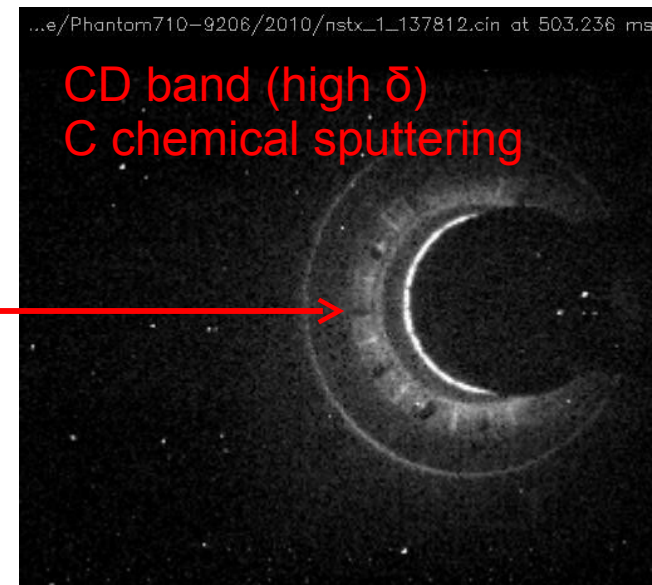


# Limited lifetime of coatings and increase in divertor $T_e$ can partially balance the expected reduction in carbon sputtering

- Reduction in carbon sputtering expected from lithium coverage can be partially balanced by:
  - Limited lifetime of coatings (see previous slides)
  - Transition to sheath limited regime at OSP: higher  $T_e$  can lead to higher physical sputtering

In addition:

- Typical surface roughness for unpolished graphite ~1-10 micron  $\gg$  than coating thickness
- Morphology of evaporated coatings presently not clear, can have effect on carbon sputtering
- Effect of leading edges in graphite tiles can be important for carbon sputtering
- Carbon sputtered from main wall can migrate and re-deposit in lower divertor, ionize and appear as a divertor source



# Summary of carbon and lithium sputtering observation in NSTX divertor

Considering the large uncertainties in this analysis:

- Evidence for lithium temperature dependent sputtering
- ~Consistent with published Li sputtering (IIAX, DIII-D)
- Evidence for toroidal asymmetry of influxes and coatings
- Evidence for deterioration of coatings
  
- Moderate reduction of carbon sputtering if compared to boronized graphite PFCs
- Still significant carbon sputtering observed
- Need to look at role of carbon chemical sputtering
- Work underway to investigate role of prompt re-deposition for Li vs C