# Surface chemistry analysis of NSTX private flux region and upper vessel tiles

C.N. Taylor, K.E. Luitjohan, B. Heim, J.P. Allain School of Nuclear Engineering, Purdue University, West Lafayette, IN 47907

C.H. Skinner, H.W. Kugel, R. Kaita, R. Maingi Princeton Plasma Physics Laboratory, Princeton, NJ 08543



53<sup>rd</sup> Annual meeting of the American Physical Society – Division of Plasma Physics Salt Lake City, UT, November 14-18, 2011



### **Outline**

- Background on NSTX tiles
- Offline control experiments
- NSTX tile analysis
  - Cleaning process on various tiles
  - NSTX tile deuterium pumping locations
  - Effect of high-temperature annealing
  - Lithium 'thickness' dependence
- Conclusions



# **Background on NSTX tiles**

- Four 2008 NSTX tiles were used in analysis.
  - Tiles passivate upon air exposure
- Many plasma configurations were used throughout the campaign.
  - Strikepoint locations
  - Wall conditioning (boronization, glow discharge, lithiumization)
- Brief procedure:
  - X-ray photoelectron spectroscopy (XPS) to examine surface chemistry
  - Clean with 1 keV Ar sputtering/annealing







# **Background on NSTX tiles**

- Four 2008 NSTX tiles were used in analysis.
  - Tiles passivate upon air exposure
- Many plasma configurations were used throughout the campaign.
  - Strikepoint locations
  - Wall conditioning (boronization, glow discharge, lithiumization)
- Brief procedure:
  - X-ray photoelectron spectroscopy (XPS) to examine surface chemistry
  - Clean with 1 keV Ar sputtering/annealing



Can we identify differences in tile surface chemistry that will tell us something about the plasma surface interactions?





# Offline controlled experiments

...Allow us to identify deuterium related interactions



The formation of new peaks or a reproducible peak shift is an indication of a chemical change.

C.N. Taylor | APS-DPP, Salt Lake City | 11/14/2011

C.N. Taylor, B. Heim, J.P. Allain, Journal of Applied Physics 109, 053306 (2011)

5

RSSEL





#### **Cleaning process on various tiles** As is spectra are inconsistent

Consider the "as is" spectra (black).

- As is spectra are not consistent from tile to tile.
- Spectra uncharacteristic of passivated Li-graphite, graphite, or lithium.
- Unusual broad XPS peaks

7

RSSEL



-- As is

C.N. Taylor, et al., Nucl. Fusion, submitted 2011.

#### Cleaning process on various tiles **Sputter cleaning recovers "most" locations**

Consider the "Ar sputtered" spectra (red).

Sputter cleaning (red peaks) does not 'repair' broad spectra for all samples



-- As is

8

C.N. Taylor, et al., Nucl. Fusion, submitted 2011.

**RSSEL** 

#### **Cleaning process on various tiles** Sputtering and heating repairs *all* samples

Consider the "Annealed" spectra (blue).

 Heating to ~550°C (blue peaks) transforms and 'repairs' broad spectra, revealing Li-O-D and Li-C-D peaks characteristic of deuterium retention.



-- As is

C.N. Taylor, et al., Nucl. Fusion, submitted 2011.

**RSSEL** 



## **NSTX tile analysis**

- So what do these peaks mean?
- Reminder:
  - From controlled studies, peaks have been identified that are associated with deuterium retention.
    - Li-O (529.5 eV) is converted to Li-O-D (532.9 eV)
    - Li-C-D forms at 291.2 eV



## **Deuterium saturation of lithiated graphite**

How is saturation calculated?





RSSEL

#### Effect of high-temperature annealing Heating to ~850°C desorbs D-related interactions

Tile A235-021 0 1s As-is sample 7hr Sputter cleaning of sample A1 revealed Ar sputtering (1hr unless noted) Li-O-D and Li-C-D peaks (shown in red) Ar sputtering (3hr) Sample annealing which are characteristic of deuterium Tile A413-017 retention. Heating to ~850°C rids sample of all (a.u.) known deuterium-related bonds. A13 Intensity Tile A408-002 Lithium remains on the sample surface even after excessive heating. σ A5 2hr Li-C-D Li CO, Li-D-X C1s Li 1s Tile A408-002 Tile A401-001 300 58 56 54 52 50 295 290 285 280 68 66 64 62 60 **Binding Energy (eV) Binding Energy (eV)** Tile A401-001 Experiments involving desorption 534 of deuterium from lithiated 544 542 540 538 536 532 530 528 Binding Energy (eV) graphite should heat to ~850°C. C.N. Taylor, et al., Nucl. Fusion, submitted 2011. PURDUE 14 C.N. Taylor | APS-DPP, Salt Lake City | 11/14/2011 ENGINEERING RSSEL

Li-O-D

Li-0

#### Lithium 'thickness' dependence Upper vessel tile has large lithium coverage gradient





C.N. Taylor | APS-DPP, Salt Lake City | 11/14/2011

ENGINEERING

Normalized Intensity (a.u.)

RSSEL

16



Normalized Intensity (a.u.)



#### Lithium 'thickness' dependence **Control experiments find minimum Li threshold**



## Conclusions

20

- Procedures for recovery of pre-passivated surface show that Ar sputter cleaning and heating is capable of uncovering peaks associated with deuterium retention.
  - To avoid excessive difficulty in resuming plasma operations after a mid-campaign vent (where mild passivation may occur), perform glow discharge *and* heating.
- Regions closest to lower divertor corner have unused deuterium pumping capability remaining while other locations are saturated with deuterium and become deuterium recycling sources.
  - Lithium conditioning should focus on high recycling regions.
- Excessive heating (~850°C) effectively depletes sample of deuterium related interactions observed in O 1s and C 1s regions.
  - Thermal desorption experiments with deuterated Li-graphite should heat to ~850°C to ensure thorough desorption.
- A minimum lithium threshold is *highly sensitive to surface morphology* and is found between 50-500 nm.



